

Advances in N-doped Diamond Thin Films grown by Microwave Plasma CVD

MICHAEL P. BRADLEY

PROFESSOR, PHYSICS & ENGINEERING PHYSICS

UNIVERSITY OF SASKATCHEWAN



UNIVERSITY OF
SASKATCHEWAN

Diamond Growth

Naturally occurring diamonds are formed in the mantle of the Earth, under conditions of high temperature and pressure

In the laboratory, diamond films can be grown using microwave plasma chemical vapour deposition (MPCVD) using a feedstock of methane gas with excess H_2

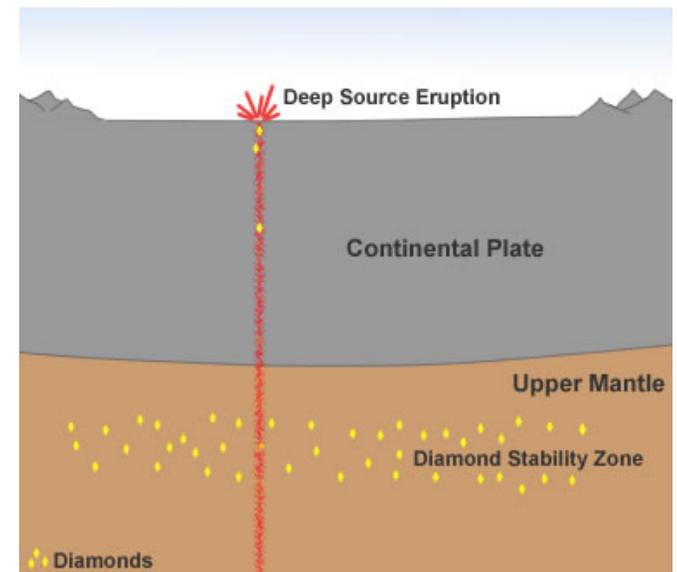
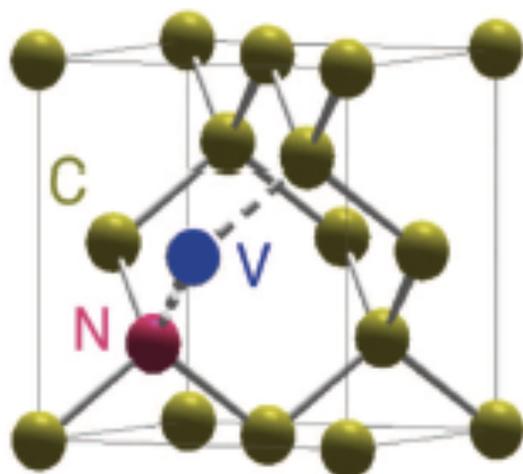


Figure credit: geology.com

Applications of Diamond Films

- ▶ Beyond the gemstone market, diamond films have a number of applications (hard coatings, transparent windows, etc.)
- ▶ Diamond is also a wide bandgap semiconductor and can also be doped
- ▶ a particularly interesting dopant is nitrogen (N)

The Diamond NV Centre



NV Centre in Diamond Schematic

[Figure Credit: D. Zheng, Ph.D. Thesis (Ecole Normale Cachan, 2010) Fig 2.3]

The NV Centre in Diamond- a special type of defect:

A substitutional nitrogen (N) *adjacent* to a vacancy (V)

Can be made in various ways:

- e-beam or Ion implantation followed by annealing at 800 C
- Nitrogen Ion implantation (introduces N and creates vacancies)
- Direct deposition of N-doped diamond films via CVD

Two NV Centre Charge States:

NV⁰, with a zero-phonon line at 575 nm

&

Negatively charged NV⁻, with a zero-phonon line at 637 nm

Magnetometry using the NV- Centre

The NV- centre acts like a localized "artificial atom" with narrow atomic-like energy states and a magnetic fine structure

→ Sub-nanotesla precision measurement of magnetic fields using NV centres is possible. This has many interesting sensor applications

- ▶ Geomagnetism: currently SQUIDS are the best sensors, but they require bulky cryogenic systems
- ▶ Electric Power Systems, Biomagnetism (pT required)
- ▶ Diamond NV centres are also possible Quantum Information Qbits

Microwave Plasma-Enhanced CVD for diamond film growth



**Microwave plasma
Regime**

Frequency 2.45 GHz

**Power: 500 - 800 Watt
Pressure: 23 - 100 Torr**

**Feed gas ratio ~ 99% H₂/1% CH₄
(excess hydrogen)**

Film growth rate:

~ 0.1 – 1 $\mu\text{m}/\text{hour}$

1. Synthesis of NV Centres in polycrystalline diamonds at low pressure

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- ▶ Polycrystalline diamond (PCD) films were deposited on a 3 mm x 3 mm x 0.5 mm
- ▶ P-type boron doped silicon substrate of
- ▶ resistivity 5Ωcm (Silicon Material Inc.) in a 2.45 GHz MPCVD reactor (Plasmionique Inc.).

- ▶ The silicon substrate has a $\langle 100 \rangle$ orientation.
- ▶ Prior to growth, the silicon substrates were nucleated by treating the surface with a mixture of diamond powder and ethyl-alcohol in an ultrasonic bath for 30 min.

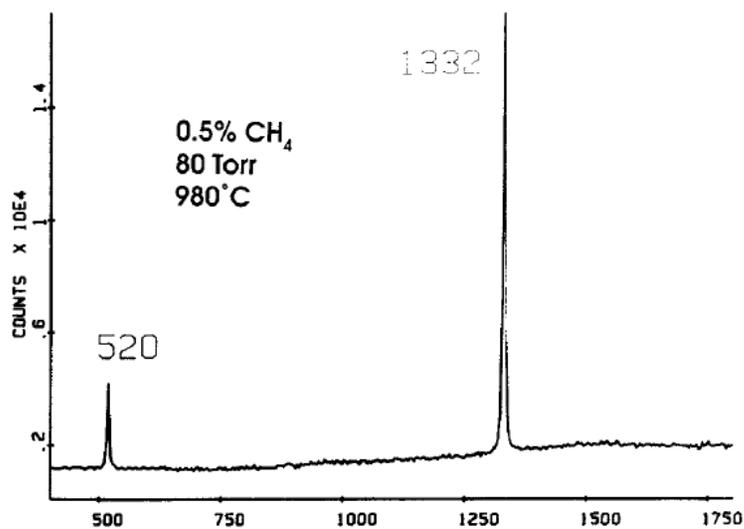
I. Synthesis of NV Centres in polycrystalline diamonds at low pressure

Diamond deposition was carried out at 800 W microwave power.
 Film growth in large excess of H₂ and 30 Torr (4 kPa) process pressure

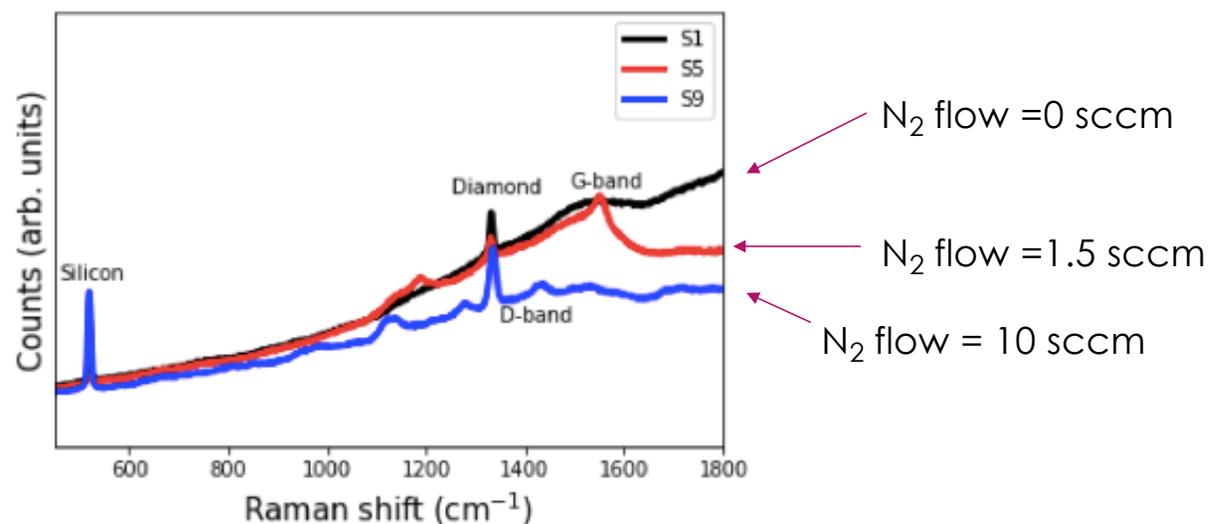
Group	Sample	H ₂ -CH ₄ -N ₂ flow (sccm)	Growth time + Doping time (hrs)	
1	S1	100-1-0	14 + 0	} 11 ± 0.1 μm
	S5	100-1-1.5	13 + 1	
	S9	100-1-10	13 + 1	
2	S2	200-1-0	14 + 0	}
	S6	200-1-2	12 + 2	
	S10	200-1-20	12 + 2	
3	S3	300-0.75-0	14 + 0	}
	S7	300-0.75-3	13 + 3	
	S11	300-0.75-30	11 + 3	
4	S4	400-0.5-0	14 + 0	} 2.5 ± 0.07 μm
	S8	400-0.5-4	10 + 4	
	S12	400-0.5-40	10 + 4	

H.A. Ejalonibu, G.E. Sarty, M.P. Bradley, *Journal of Materials Science: Materials in Electronics* (2019) 30:10369–10382

Signature of Diamond: Raman line at 1332 cm^{-1}



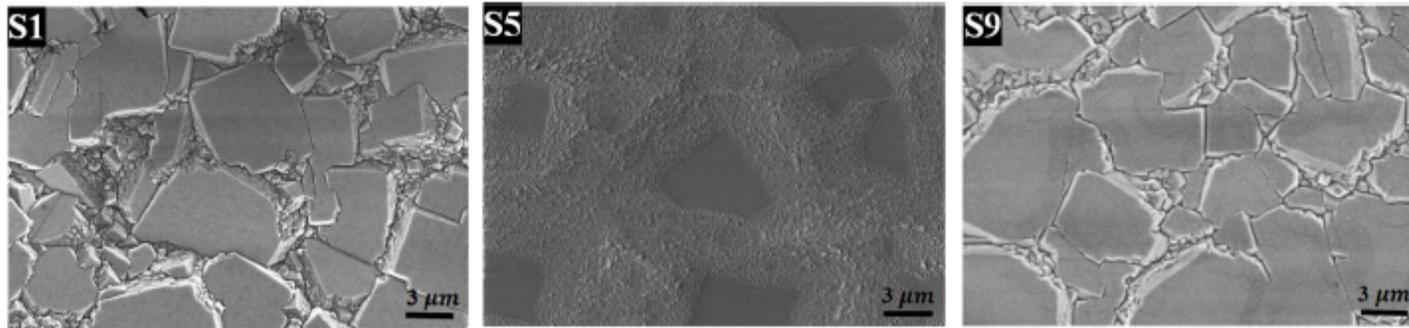
Raman Spectra of Films in Group 1 (100 sccm H₂/1.5 sccm CH₄)



Raman Peak

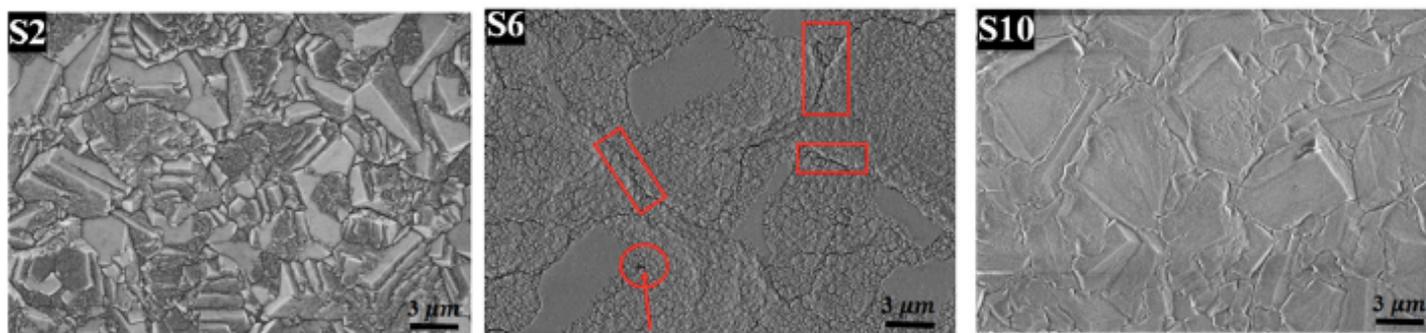
Group	Sample	FWHM (D) (cm ⁻¹)	Center of Dia- mond peak (cm ⁻¹)	Color
1	S1	8.232	1331.2	Grey
	S5	7.695	1331.7	Lustrous grey
	S9	7.792	1332.0	Lustrous grey
2	S2	10.989	1331.3	Grey
	S6	11.268	1331.9	Lustrous grey
	S10	8.875	1331.9	Lustrous grey
3	S3	8.201	1332.0	Grey
	S7	13.320	1331.2	Grey
	S11	14.662	1332.5	Black
4	S4	8.586	1331.5	Grey
	S8	8.752	1331.4	Dark grey
	S12	15.463	1333.8	Black

Surface Morphology Group 1 Samples



Lowest H₂ flow case: 100 sccm

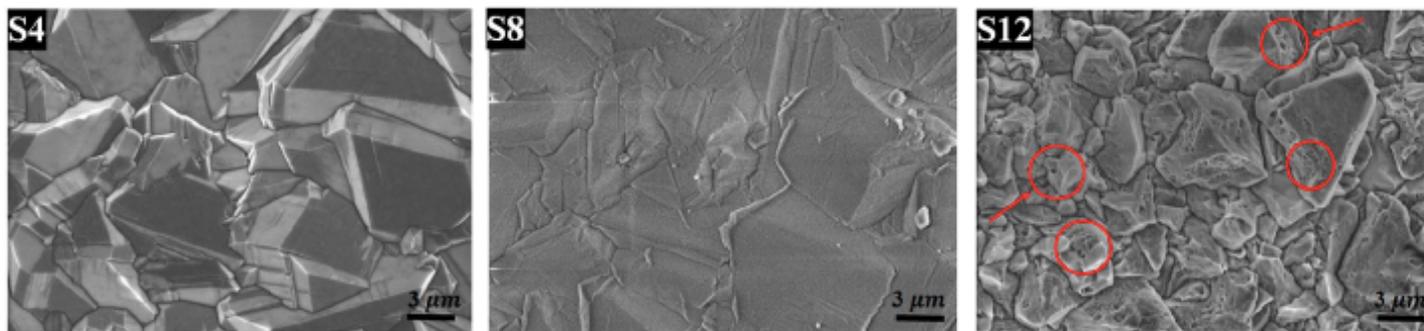
Surface Morphology Group 2 Samples



Note presence of cracks

Higher H₂ flow case: 200 sccm

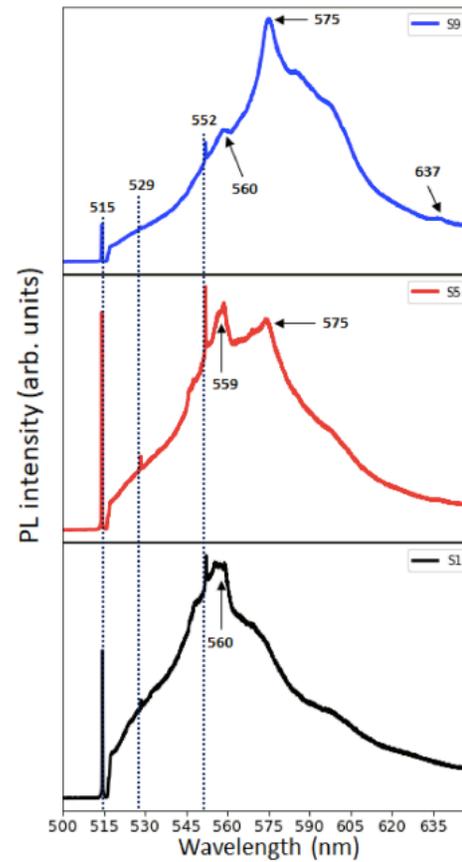
Surface Morphology Group 4 Samples



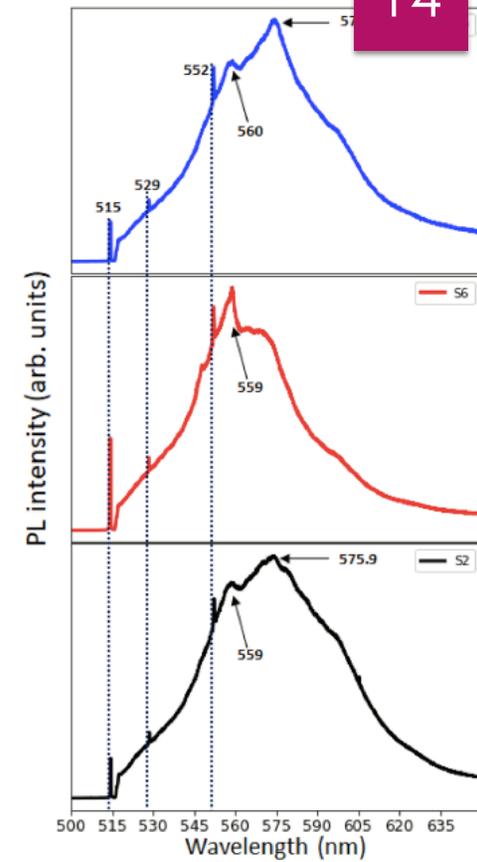
Highest H₂ flow case: 400 sccm

Photoluminescence Spectra

- ▶ 515 nm line: Ar ion laser
- ▶ 575 nm: NV⁰
- ▶ 637 nm: NV⁻



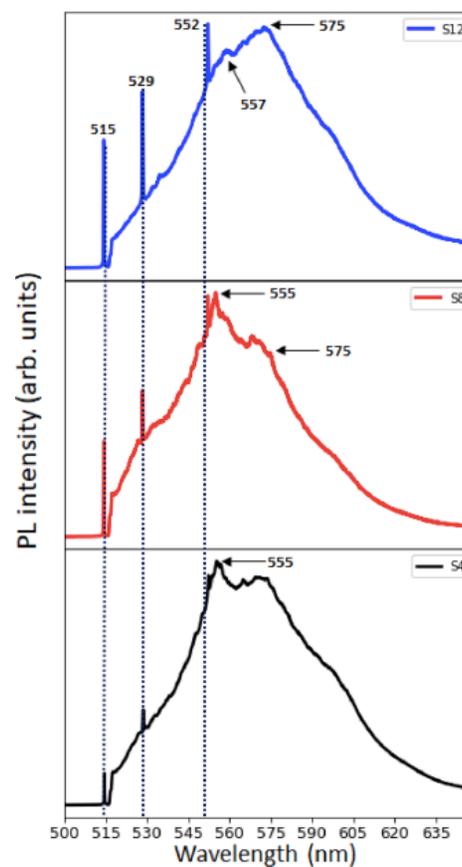
Group 1



Group 2

Photoluminescence Spectra

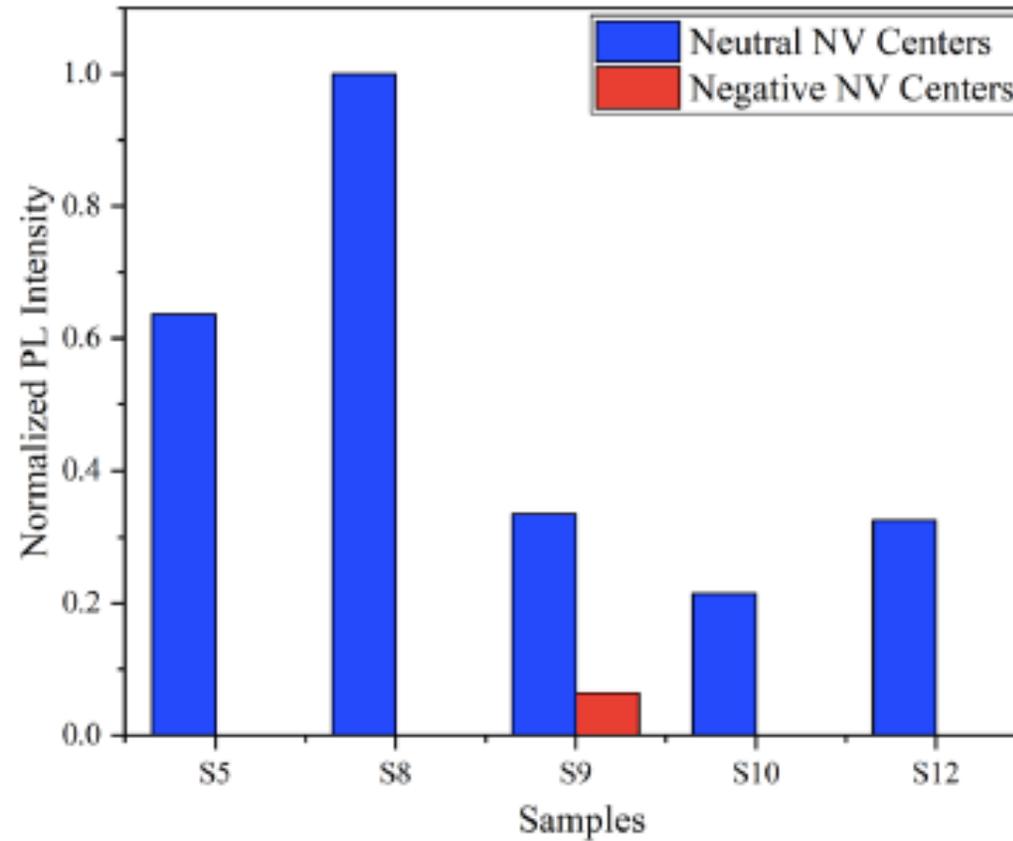
- ▶ 515 nm line: Ar ion laser
- ▶ 575 nm: NV⁰
- ▶ 637 nm: NV⁻



Group 4

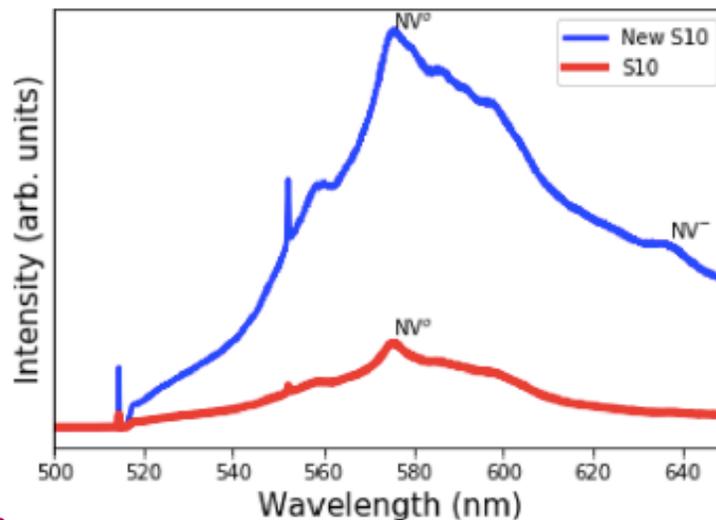


Only sample S9 exhibited a (small) NV⁻ 637 nm PL peak
S9 growth conditions: 100 -1-10 H₂-CH₄-N₂ flow (sccm)



Getting an NV- Centre: operate at higher pressure

Sample "New S10": Higher process pressure 50 Torr (6.7 kPa) process pressure & microwave power (900 W) with same flow rates as sample S10 (200 sccm H₂, 1 sccm of CH₄, and 20 sccm of N₂).



637 nm NV-
Photoluminescence peak

II. Step-wise surface nitrogen doping in MPECVD grown polycrystalline diamonds

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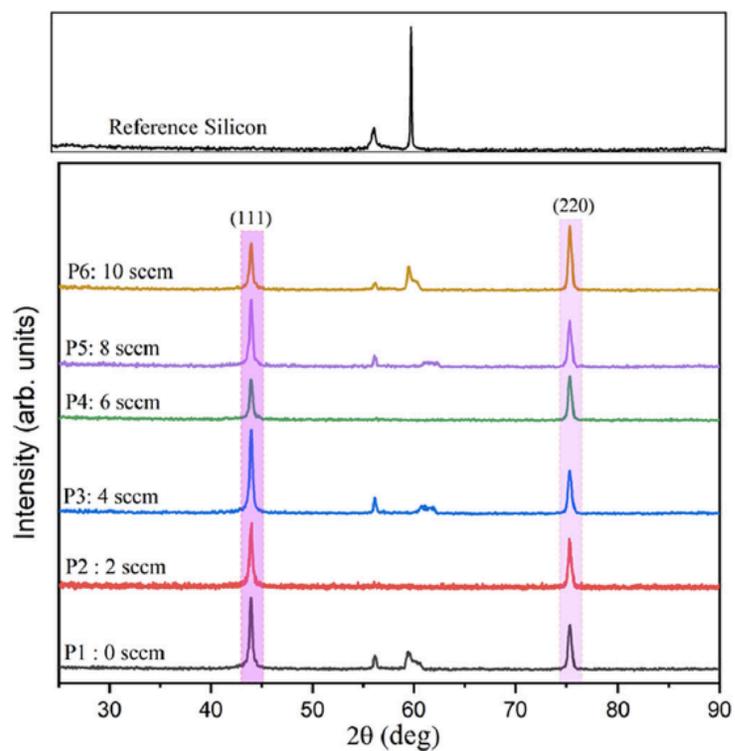
Focused on the 100 sccm H₂ / 1 sccm CH₄ flow regime & vary N₂ flow

Samples	H ₂ /CH ₄ flow (sccm)	N ₂ flow (sccm)	Growth time (h)	Doping time (h)
P1	100/ 1	0	18	0
P2	100/ 1	2	17	1
P3	100/ 1	4	17	1
P4	100/ 1	6	17	1
P5	100/ 1	8	17	1
P6	100/ 1	10	17	1

- ▶ 1. At low surface nitrogen flow rates ($[N/C] < 0.0020$), the diamond
- ▶ growth mechanism is slow enhancing secondary nucleation
- ▶ process that favours the growth of nanocrystalline grains. The nanocrystalline
- ▶ grains formed have large grain boundary density
- ▶ containing a large quantity of disordered $sp^2/a-C$ content in the
- ▶ grain boundaries.

- ▶ 2. At medium surface nitrogen flow rates ($0.0020 < [N/C]$
- ▶ < 0.0022), grain size increases which introduces some degree of
- ▶ orderliness to the sp² content in the grains and at the grain boundaries.
- ▶ The quantity of sp²/a-C phases present at this stage still prevents
- ▶ the formation of NV– centres.

- ▶ 3. At high surface nitrogen flow rates ($[N/C] > 0.0025$), the
- ▶ quantity of nano-grain sized diamond reduces except at the grain
- ▶ boundaries. The degree of disordered $sp^2/a-C$ content decreases with
- ▶ increasing grain size thereby enabling the formation of NV- centres
- ▶ on the grains. Also, the grain boundary line density is greatly reduced



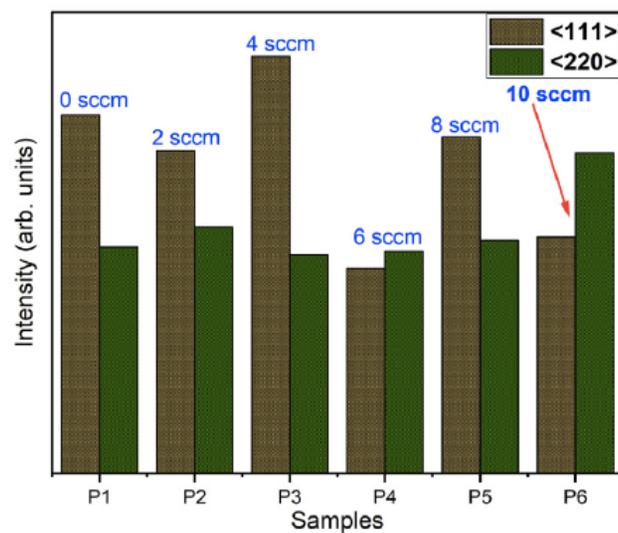
Grazing Incidence X-Ray Diffraction

Shifting the preferred diamond grain orientation by varying N₂ flow

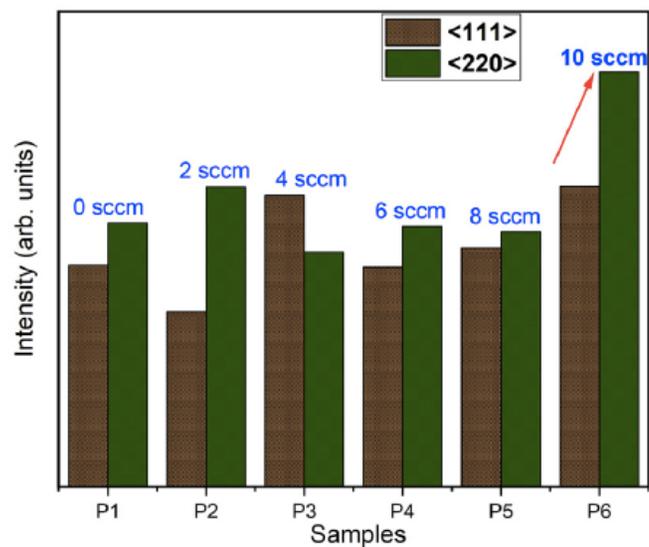
Table 4
Preferential orientation of the deposited films under varying surface nitrogen concentration.

Samples	Experiment A [$I_{(111)}/I_{(220)}$]	Experiment B [$I_{(111)}/I_{(220)}$]	Preferred Orientation
P1	1.583	0.532	Random
P2	1.313	0.584	Random
P3	1.912	1.242	$\langle 111 \rangle$
P4	0.922	0.844	$\langle 111 \rangle$
P5	1.444	0.935	Random
P6	0.738	0.724	$\langle 220 \rangle$

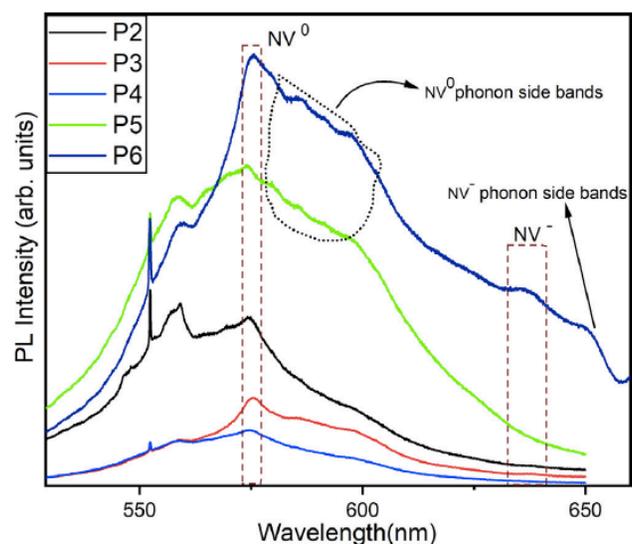
“Experiment A”



“Experiment B”

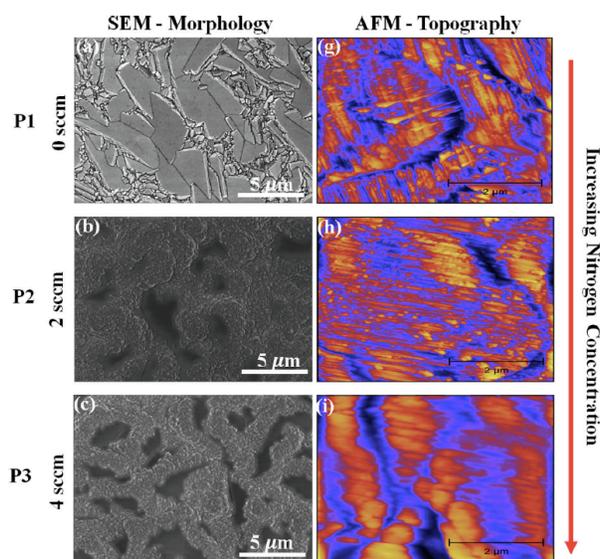


Only highest N₂ flow sample shows evidence of NV⁻ centres

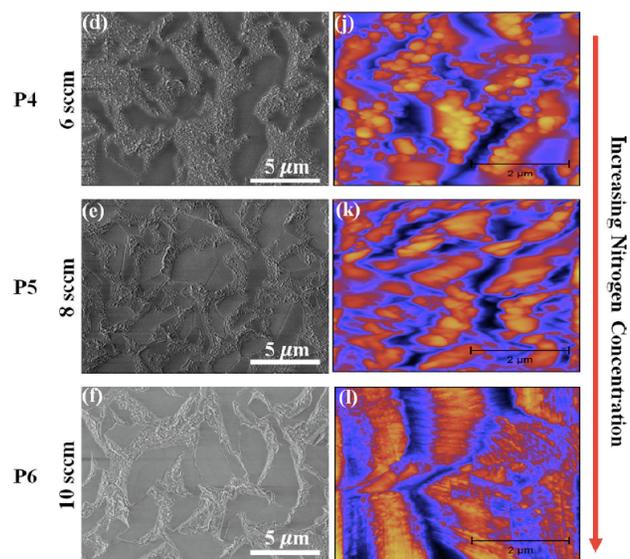


Room temperature PL emission spectra of nitrogen doped films showing the presence of NV⁰ centres in sample P2–P6 and their phonon side bands. Only sample P6 indicates the presence of NV⁻ centres.

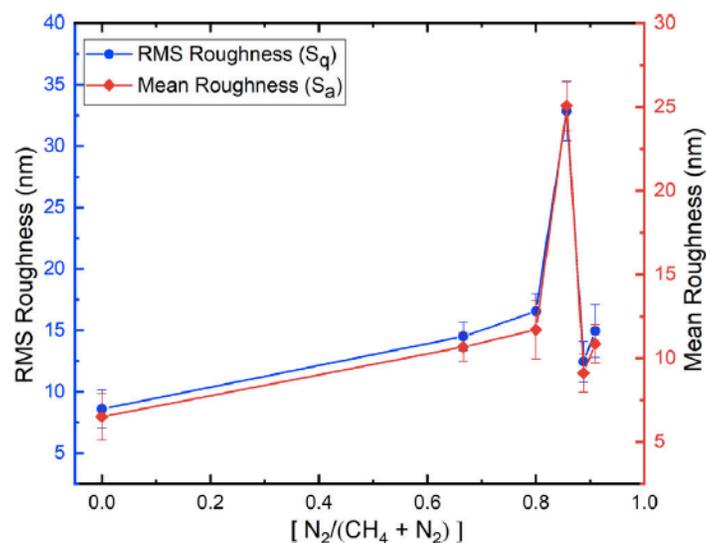
Surface Roughness: SEM & AFM Study



Surface Roughness: SEM & AFM Study



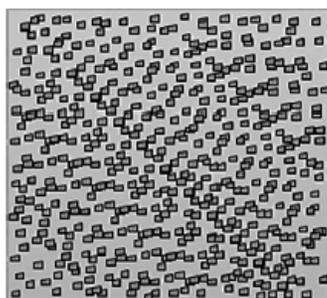
Surface Properties: drop in surface Roughness at Higher N₂ flows



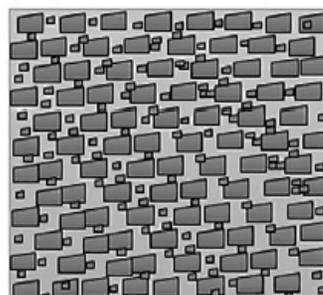
$$S_a = \frac{1}{A} \iint_A |P(x, y)| dx dy$$

$$S_q = \sqrt{\frac{1}{A} \iint_A P^2(x, y) dx dy}$$

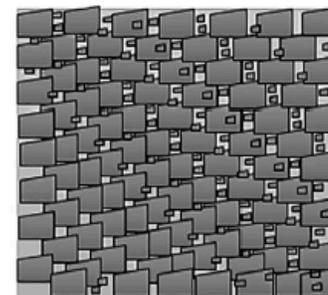
Schematic Growth Model



Low



Medium



High

Conclusions

- ▶ Growth of N-doped diamond films investigated under different growth conditions
- ▶ In general NV0 centres can be observed in the as-grown films
- ▶ Film morphological properties relevant to sensor applications
- ▶ NV- centres more difficult to achieve.
- ▶ Next step: determine plasma parameters of film growth conditions

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