

# Nanostructure of $(\text{Yb}_x \text{Y}_{1-x})_2 \text{O}_3$ Films Revealed With BF and HAADF STEM

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# Materials for High-Power Lasers

- Solid-state lasers have reached >10-kW-grade power in one or a few transverse modes.
- The maximum average power is limited by overheating and thermal fracture:
  - Need laser architecture with efficient cooling
  - Need for materials with a high thermal shock parameter

$$R_t = \frac{K_{th} (1 - \nu)}{\alpha E} S_t$$

$K_{th}$ : thermal conductivity  
 $\nu$ : Poisson coefficient,  
 $\alpha$ : thermal expansion coefficient [ $^{\circ}\text{C}-1$ ]  $E$ :  
Young Modulus [Pa],  $S_t$ : fracture limit [Pa]

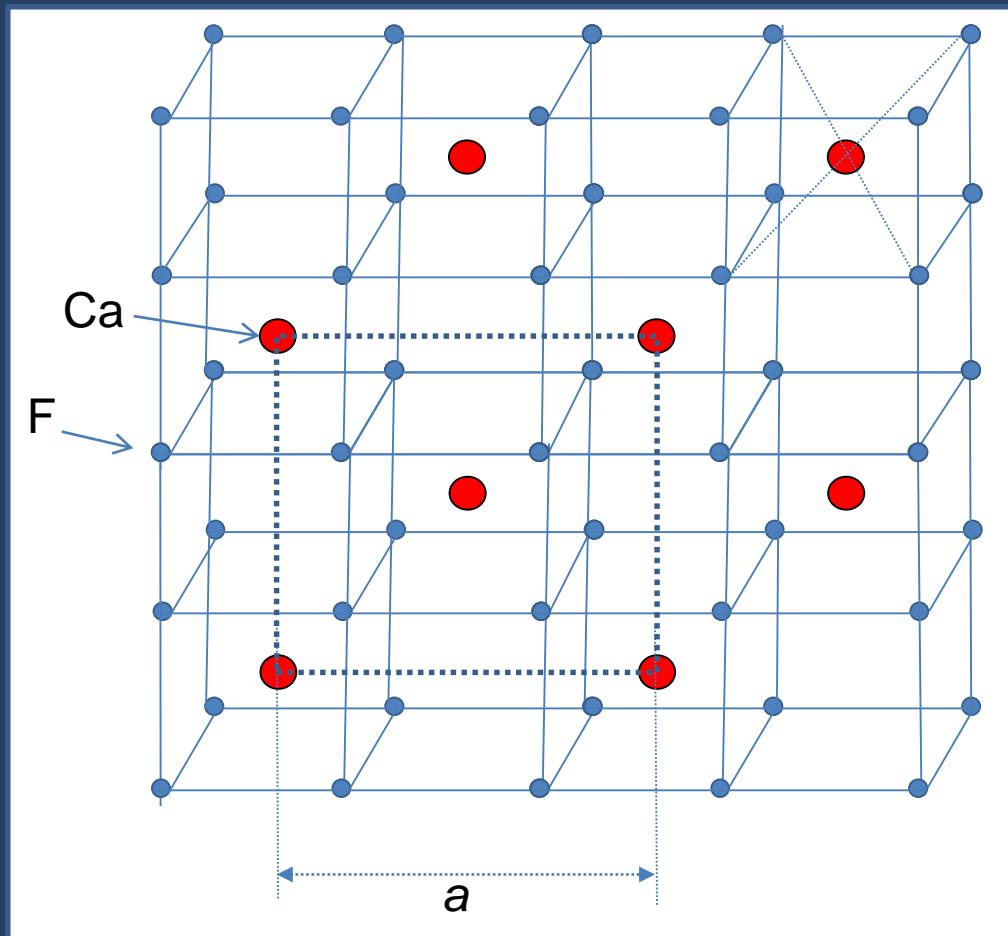
State-of-the-art crystalline material is:  $\text{Yb}^{3+}:\text{Y}_3\text{Al}_5\text{O}_{12}$

# $\text{Y}_2\text{O}_3$ : a Great Host for High-Power Lasers

- Low phonon energy ( $550 \text{ cm}^{-1}$  vs  $700 \text{ cm}^{-1}$  for YAG) => helps reduce non-radiative multiphonon relaxation;
- High density of rare-earth substitution sites (twice as YAG)
- High melting point:  $2430^\circ\text{C}$  (YAG:  $1930^\circ\text{C}$ ): Difficult to grow using traditional methods (Chrochalskii)
- Cubic material (optically isotropic): can be made transparent in the polycrystalline form using the ceramic process or thin-film deposition techniques

# Crystallographic Structure of $\text{Y}_2\text{O}_3$

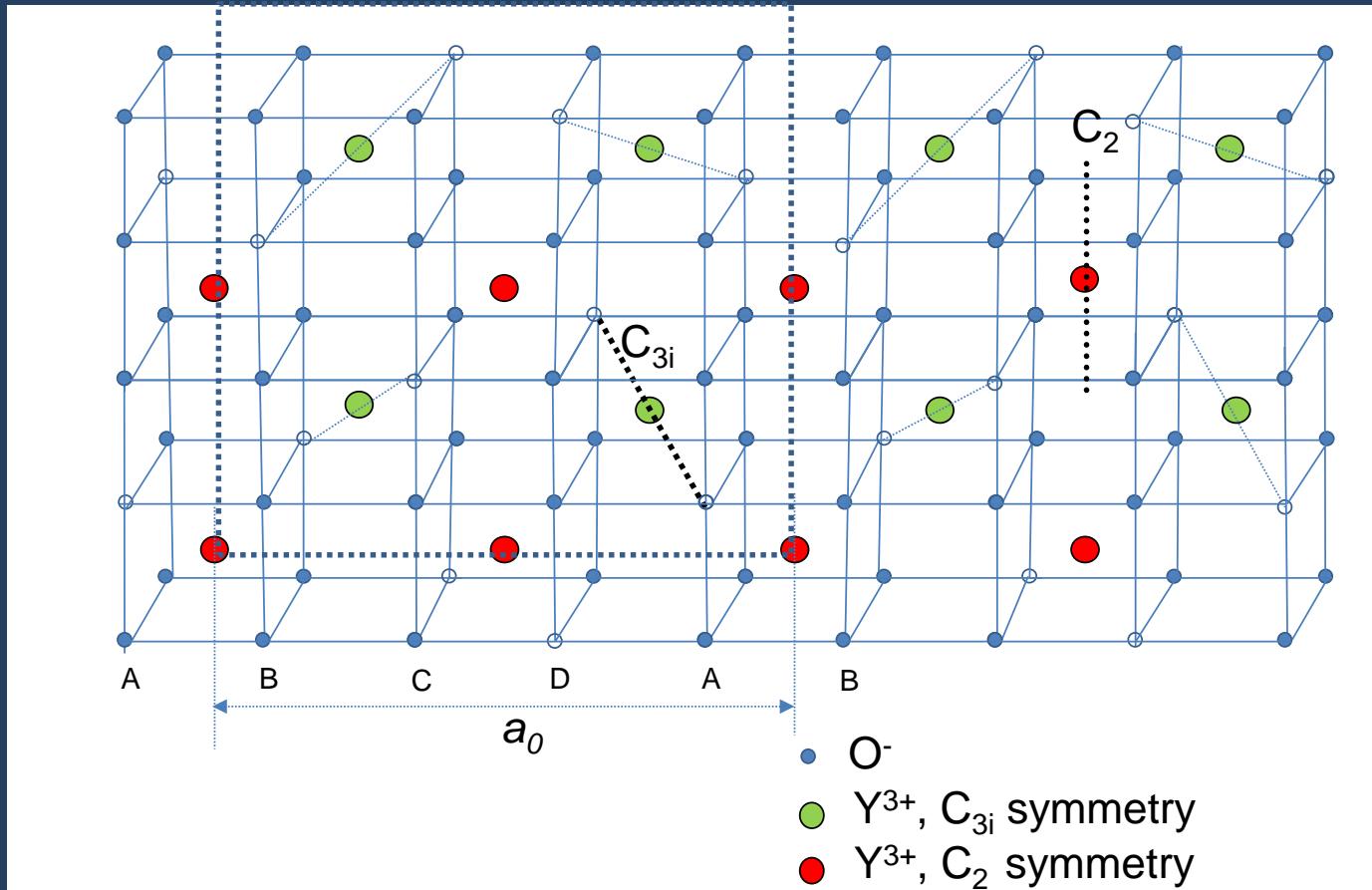
Similar to  $\text{CaF}_2$



The  $\text{CaF}_2$  lattice is face-centered-cubic (FCC).



# Crystallographic Structure of $\text{Y}_2\text{O}_3$



Similar to  $\text{CaF}_2$  but with one fourth of the  $\text{O}^{2-}$  removed

$\text{Y}_{32}\text{O}_{48}$  : 48  $\text{O}^{2-}$  sites = 64 -16 "vacancies" ; 32  $\text{Y}^{3+}$  sites=  $24\text{C}_2+8\text{C}_{3i}$

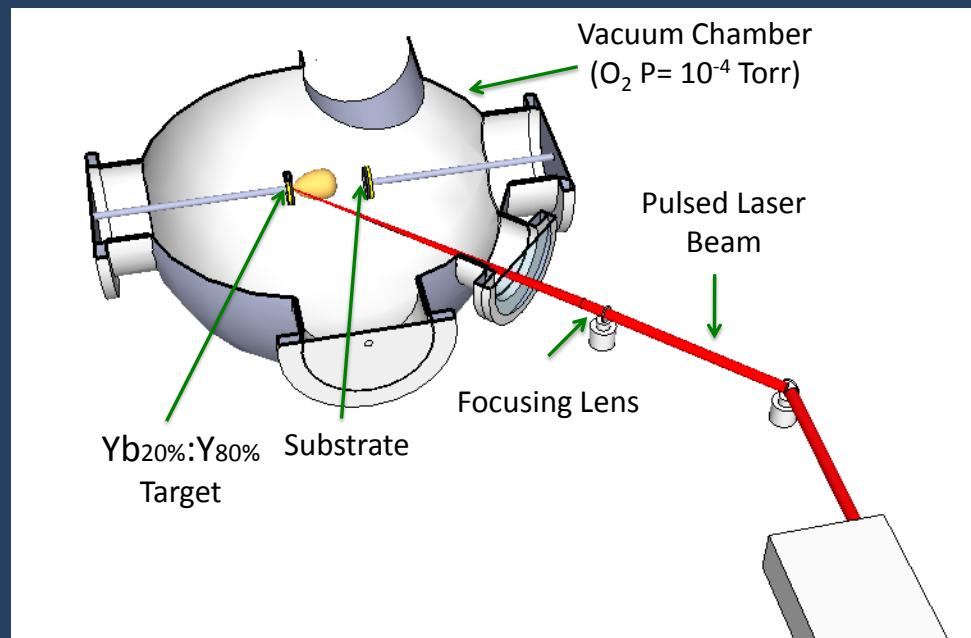
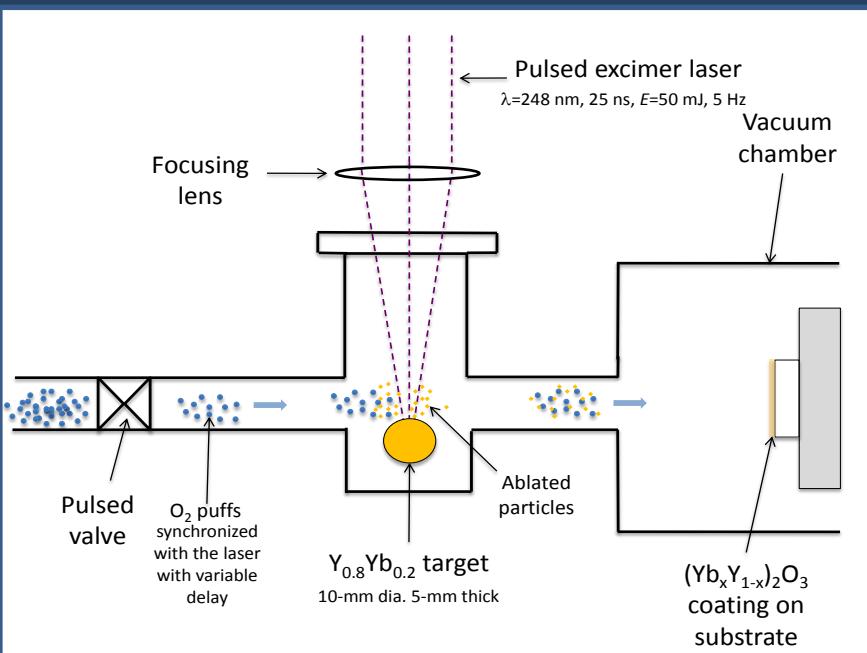
Bravais lattice: BCC;  $\text{T}_h^7$  (206)



# Thin-Film Synthesis

## Reactive cross-beam laser ablation

## Pulsed laser deposition



# STEM

- 70-nm-thick specimens prepared by focused ion beam.
- JEOL 2200FS (200 kV,  $I_p=150$  pA), probe dia.=0.12 nm, Depth of field : 10 nm

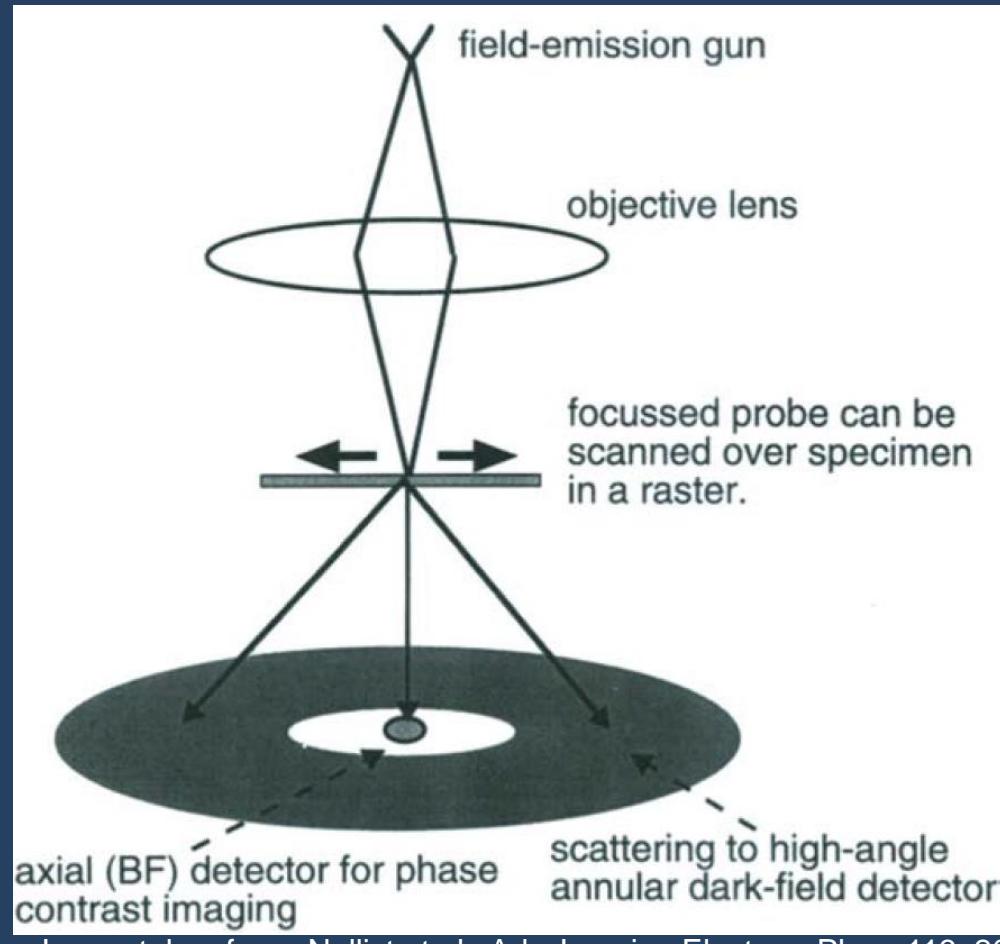
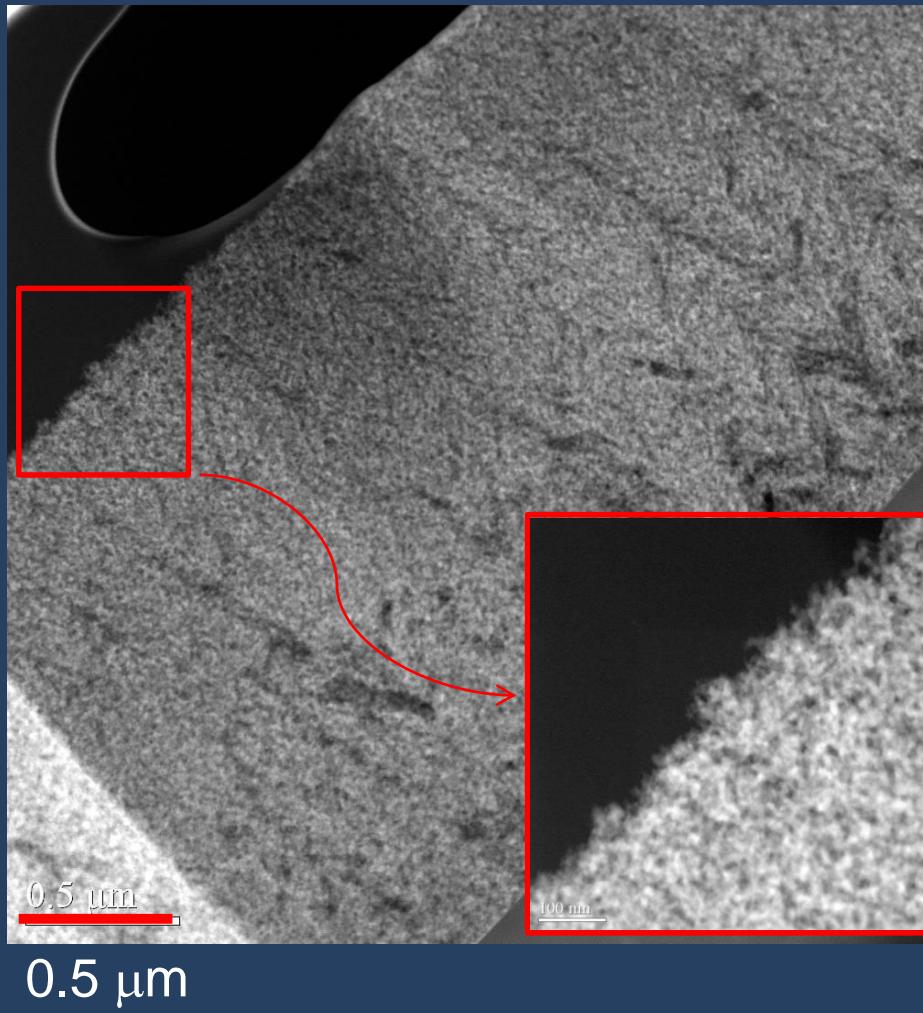


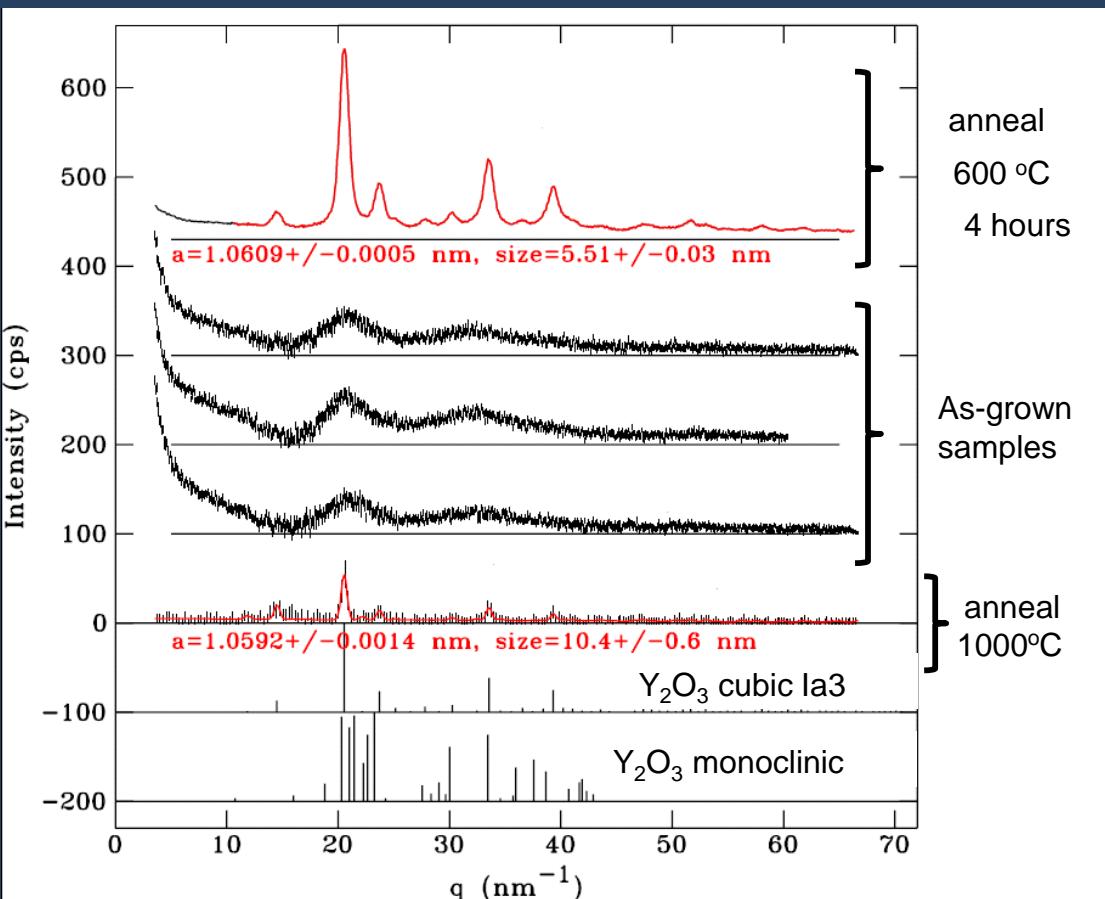
Image taken from Nellist et al., Adv. Imaging Electron. Phys. 113, 2000

- BF contrast is based on interference between diffraction orders at the detector (coherent) imaging.
- HAADF contrast :
  - Detection angle: 100 and 170 mrad.
  - Intensity proportional to  $Z^\alpha$ , where  $\alpha=1.7$
  - Higher signal for Y ( $Z=39$ ) or Yb ( $Z=70$ ) than O ( $Z=8$ ).

# STEM-HAADF Picture of As-Grown Coatings



# X-ray diffraction patterns with various annealing conditions



Electronic diffraction pattern

$10 \text{ } 1/\text{nm}$

Échelle de nombre d'onde  $q/2\pi$

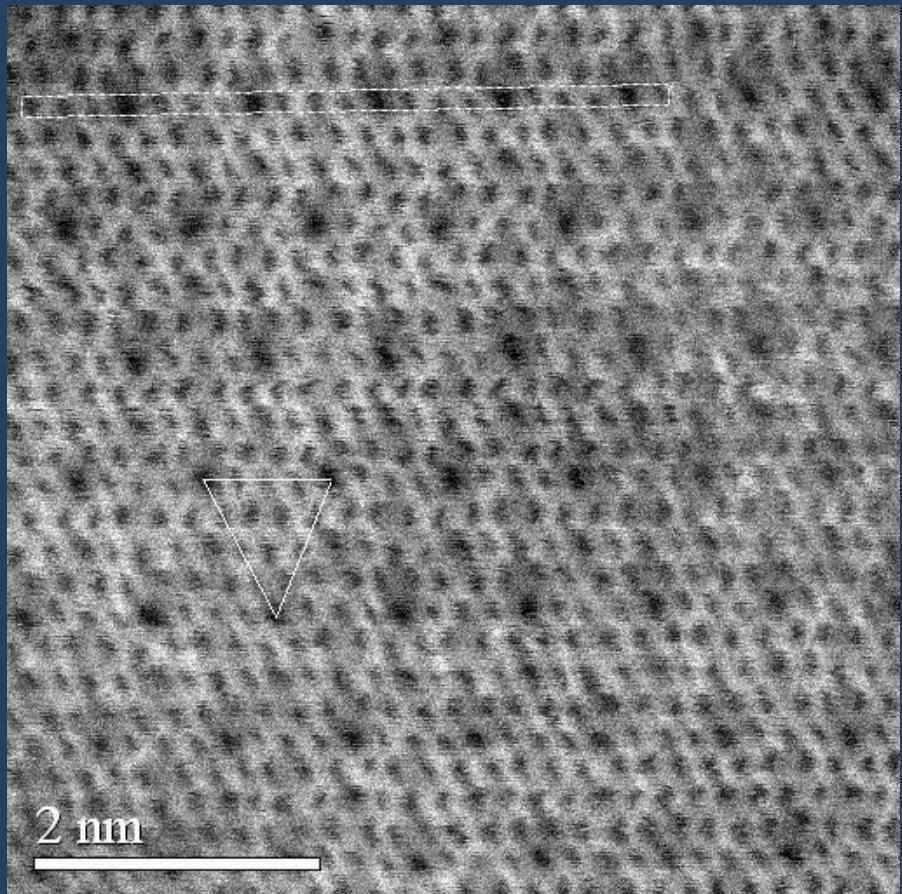
- As-grown samples: amorphous
- Annealed samples:  $\text{Y}_2\text{O}_3$ 
  - 600 °C / 4 hours =>  $L_c = 5 \text{ nm}$ ;
  - 1000 °C / 4 hours =>  $L_c = 10 \text{ nm}$ .

Measured by Ralf Bruening, Univ. Mount-Alison

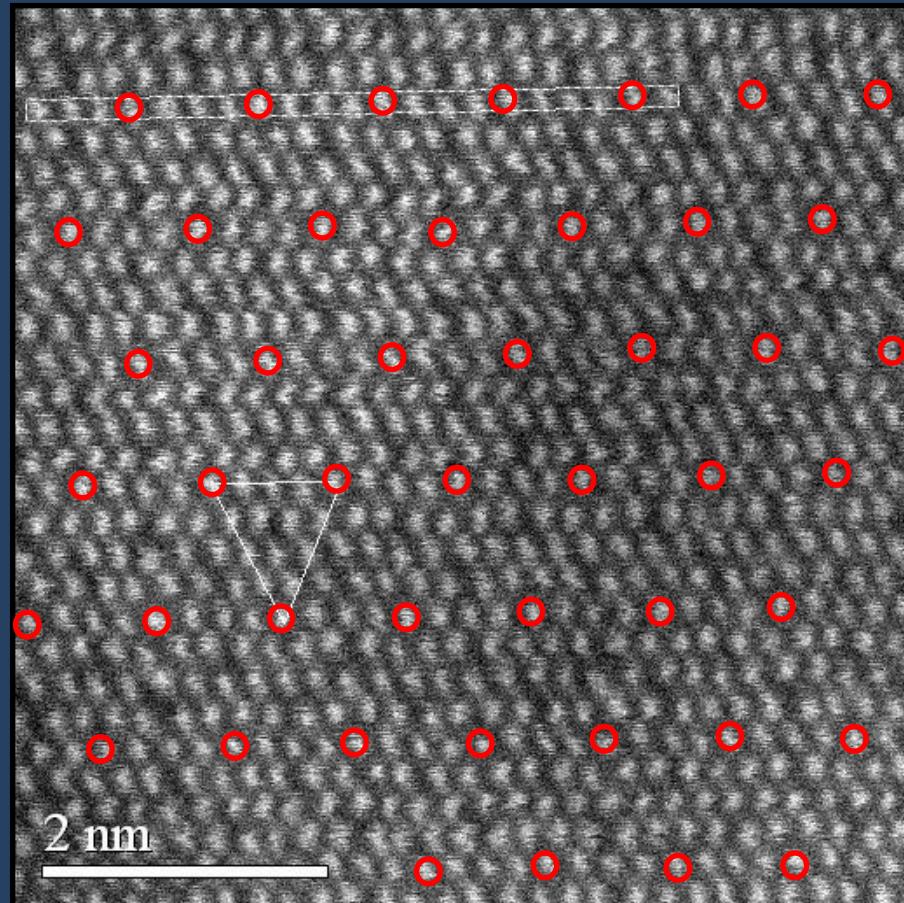


# BF and HAADF images, <111> Zone axis

Bright-Field STEM

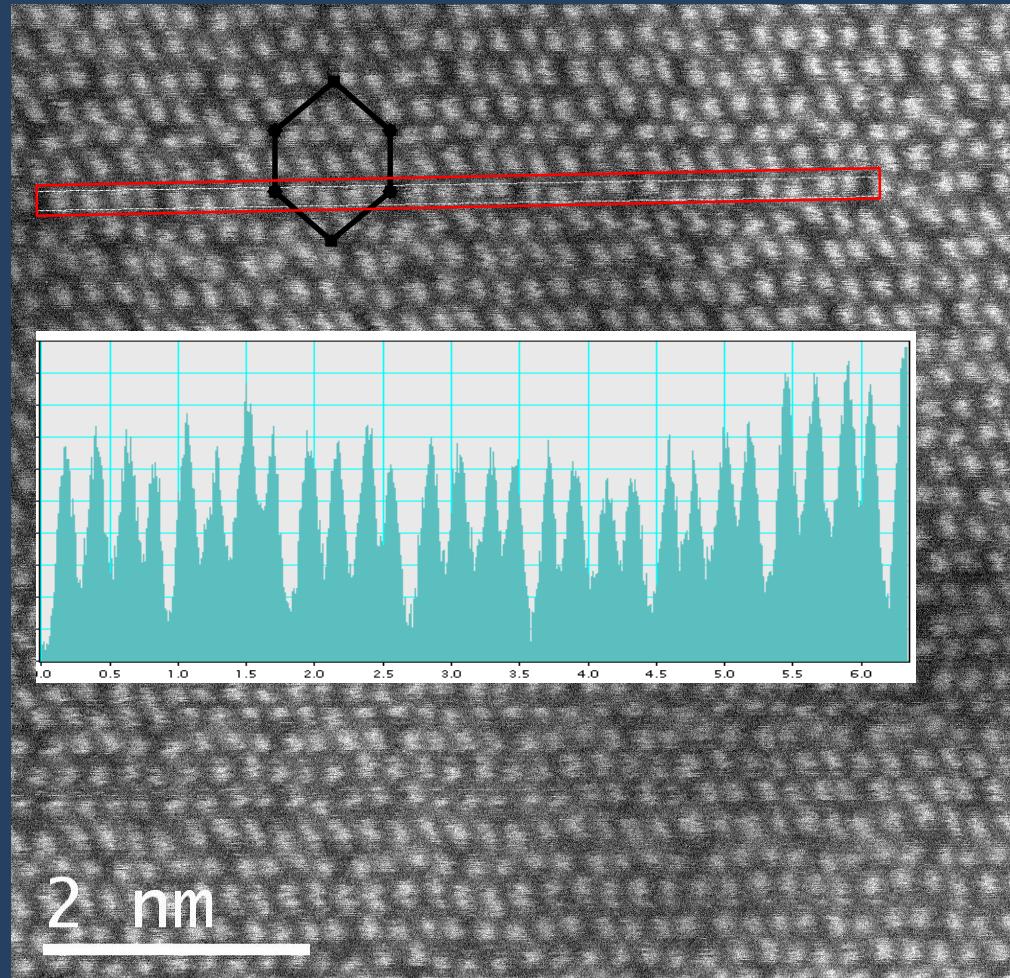


HAADF STEM



BF image suggests the presence of vacancies with long-range order, but these are absent in the HAADF image.

# Different Long-Range Order in HAADF Images



Is this the conventional structure of  $\text{Y}_2\text{O}_3$ ?

# Electron Diffraction Pattern <111>



<111>



10 1/nm

( $2p+1, 2q+1, 0$ ) peaks  
forbidden for the  $\text{Y}_2\text{O}_3$   
bixbyite structure

Different crystallographic  
structure?  
or  
Effect of secondary elastic  
scattering?

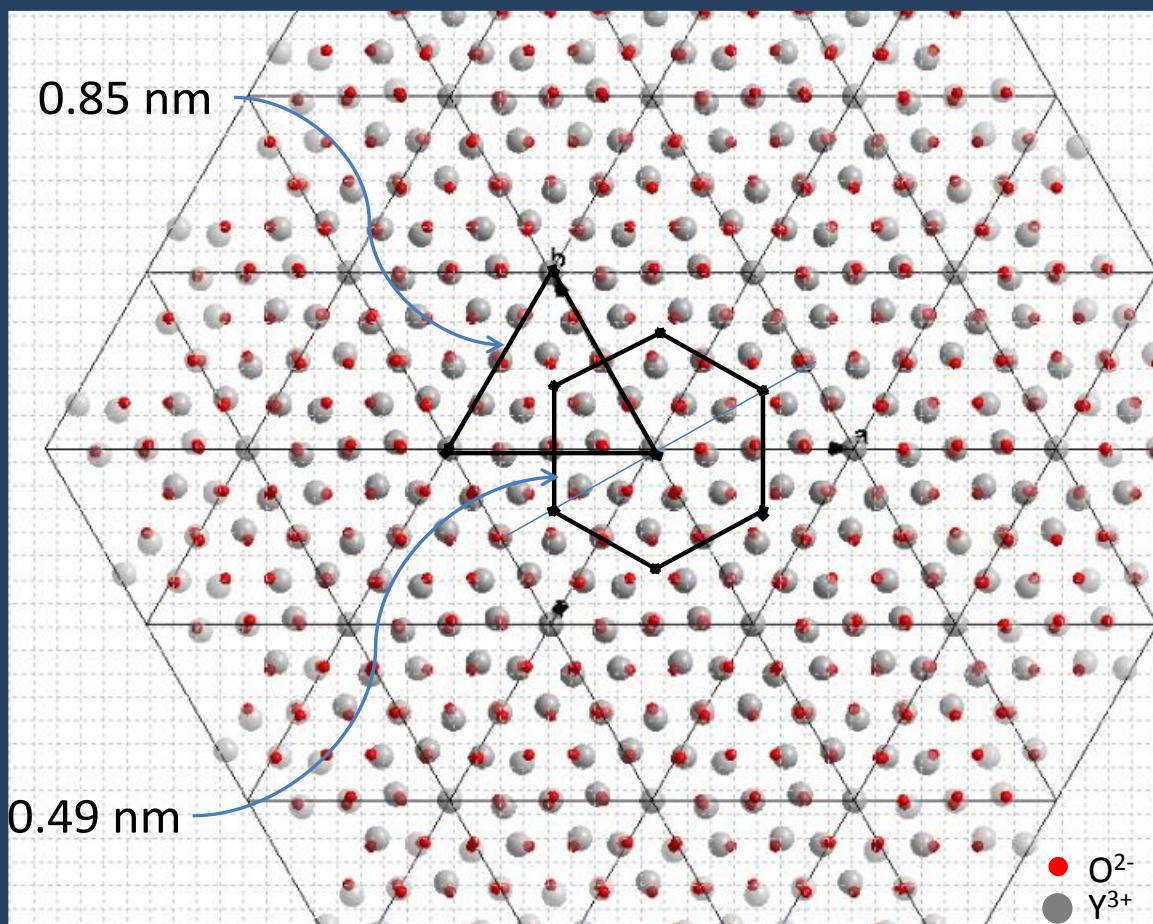
# Simulation of the $\text{Y}_2\text{O}_3$ Structure

Ion sites	Wyckoff position	Number of sites in elementary cell	Coordinate $x/a$	Coordinate $y/a$	Coordinate $z/a$
$\text{Y}^{3+}$	b (sym. $\text{C}_{3i}$ )	8	1/4	1/4	1/4
	d (sym. $\text{C}_2$ )	24	$x_A$	0	1/4
$\text{O}^{2-}$	e	48	$3/8 + x_0$	$1/8 + y_0$	$3/8 + z_0$

$$a=1.06 \text{ nm}, x_0=0.0157, y_0=0.0270, z_0=0.0054, x_A=-0.0327 \quad <111>$$

Ref: Hanic et al., Acta Cryst B40 : 76-82, 1984

- 1) Dark spots in BF coincide with the absence of oxygen ions;
- 2) Dark spots HAADF correspond a distortion of the cationic network.



# Summary

- BF and HAADF images of Yb:  $\text{Y}_2\text{O}_3$  thin films reveal complimentary informations, each revealing a different pattern of dark spots organized in a honeycomb structure:
  - HAADF images show periodic distortion of the cationic columns;  $\text{O}^{2-}$  was not detected.
  - BF images reveal the absence of oxygen ions at specific crystallographic columns;
  - Combining BF and HAADF modes makes it possible to get a comprehensive picture of the atomic structure of a material.

# Acknowledgments

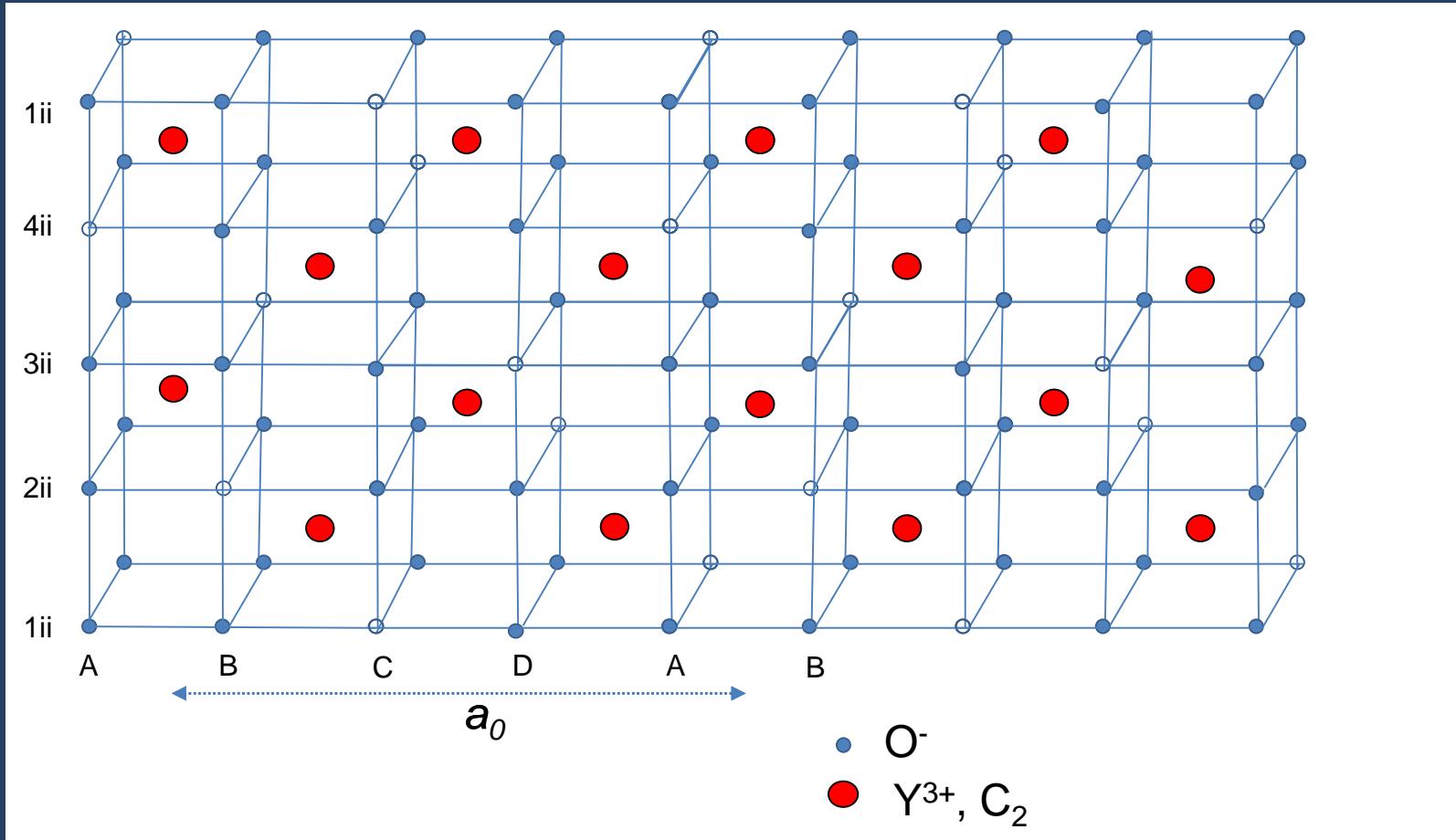
- David Troadec, CNRS, France
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- New-Brunswick Innovation Foundation (NBIF)
- Canada Foundation for Innovation (CFI)
- Faculté des études supérieures et de la recherche de l'Université de Moncton

# Crystallographic Structure of $\text{Y}_2\text{O}_3$

$\text{Y}_{32}\text{O}_{48}$

32  $\text{Y}^{3+}$  sites =  $24\text{C}_2 + 8\text{C}_{3i}$

48  $\text{O}^{2-}$  sites =  $64 - 16$  "vacancies"

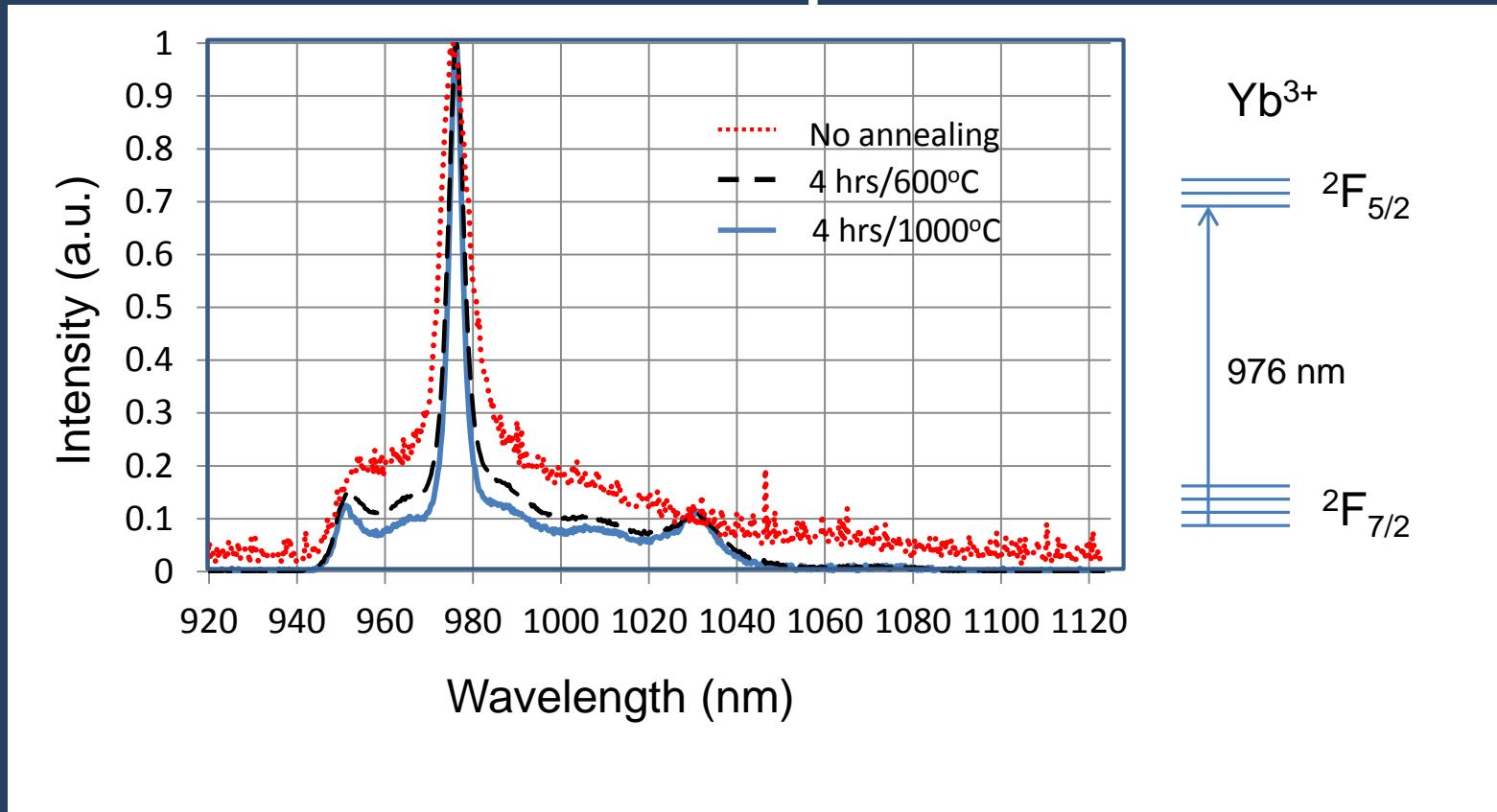


Bravais lattice: body centered cubic;  $T_h^7$  (206) space group

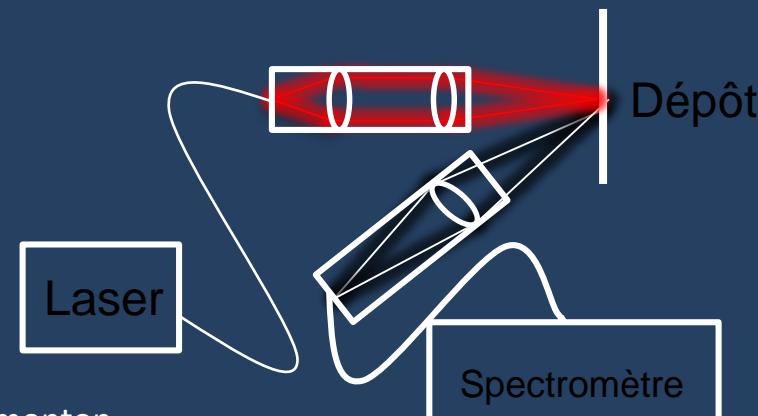


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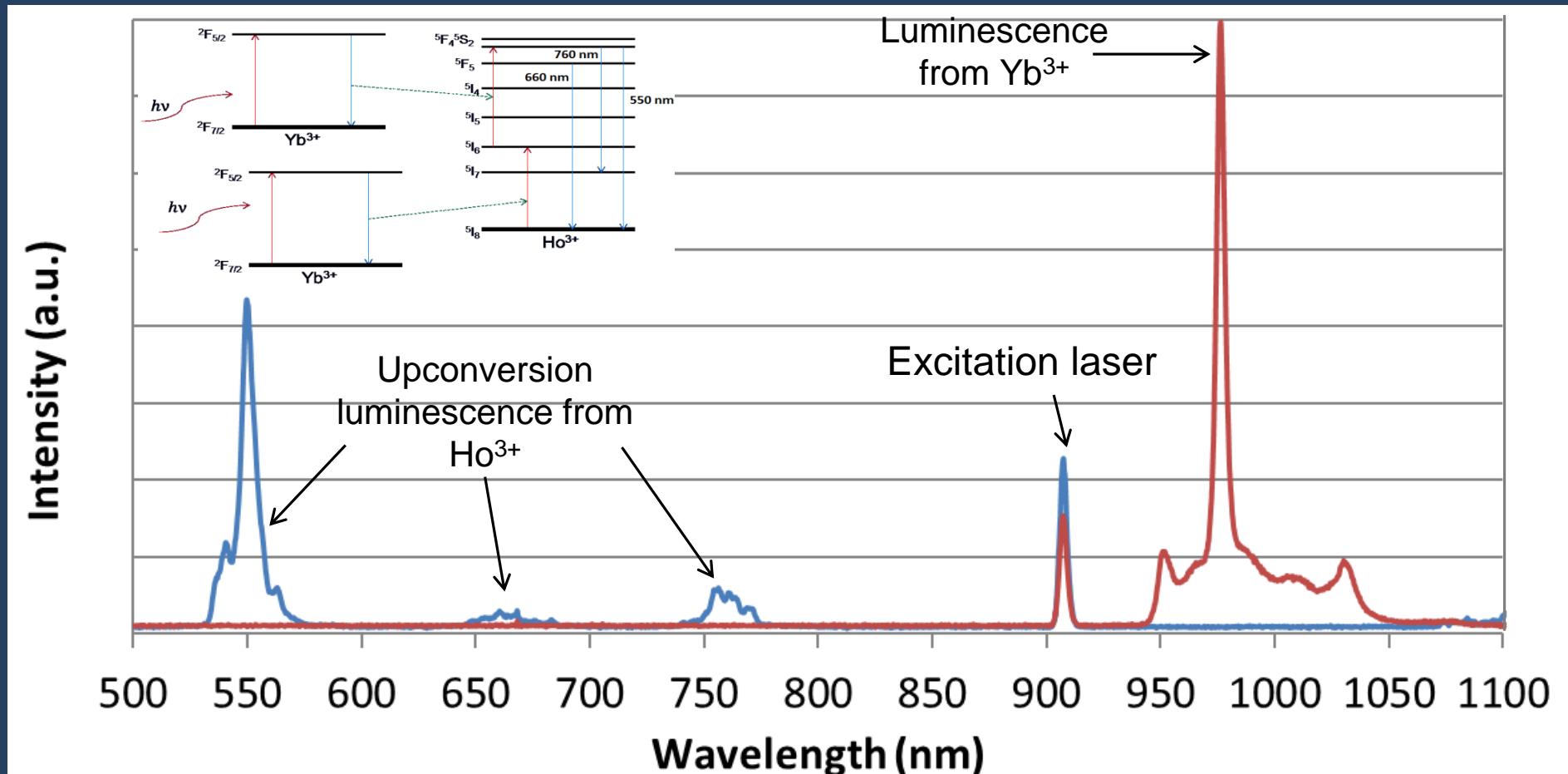
# Luminescence Spectra from $\text{Yb}^{3+}$



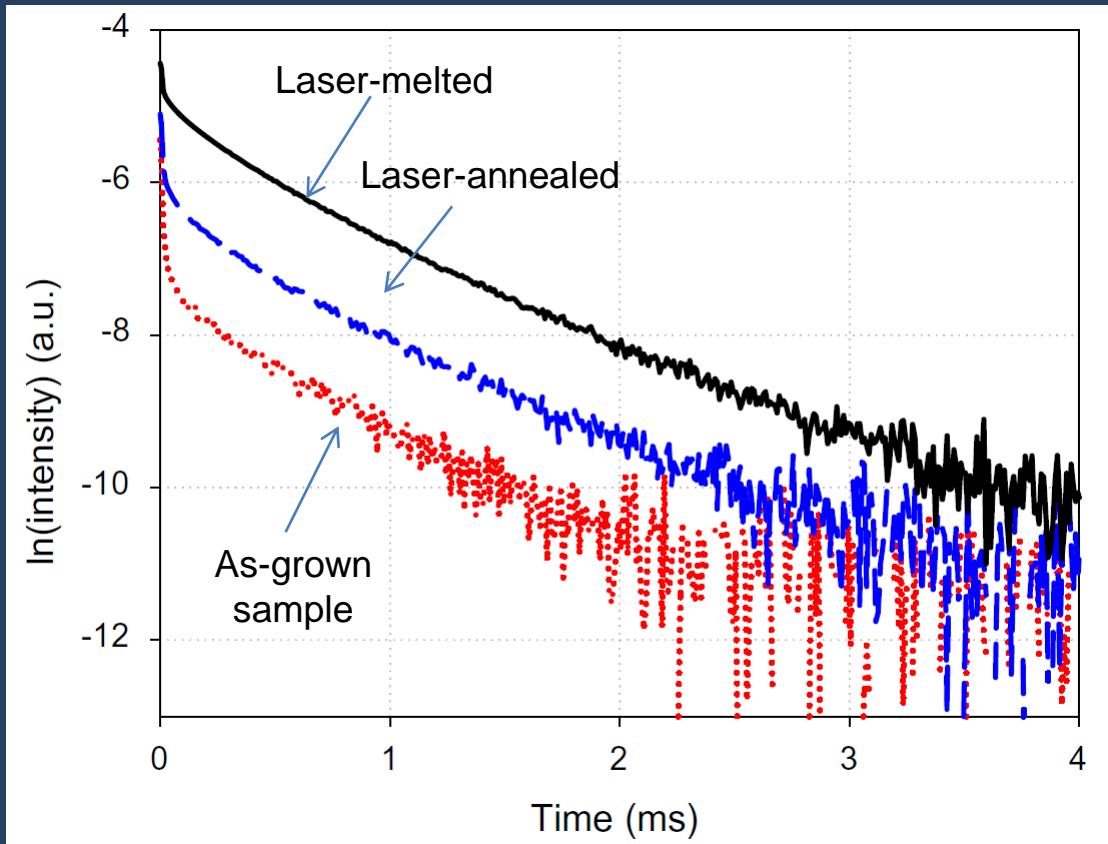
1. Significant broadening and little structure for the amorphous material => Suggests that the crystal field around  $\text{Yb}$  ions is inhomogeneous
2. Spectra narrow as the size of the crystallites increases



# Upconversion Luminescence From Other Trivalent Rare-Earth Ions



# Luminescence Decay Curves



- Non-exponential decay: can be explained by the presence of randomly distributed traps in the material (Cf. J.F. Bisson, JOSA B, 32(5). 757-766, 2015);
- Important jump in luminescence intensity after melting the material;
- Lifetime at long times in the order of :  $\tau=0.93$  ms (about the same for all cases).

